

Possible Superconductivity in Fe-Sb Based Materials: Density Functional Study of LiFeSb

Lijun Zhang¹, Alaska Subedi^{1,2}, D.J. Singh¹, and M.H. Du¹

¹*Materials Science and Technology Division, Oak Ridge National Laboratory, Oak Ridge, Tennessee 37831-6114 and*

²*Department of Physics and Astronomy, University of Tennessee, Knoxville, TN 37996*

(Dated: October 27, 2008)

We investigate the electronic and other properties of the hypothetical compound LiFeSb in relation to superconducting LiFeAs and FeSe using density functional calculations. The results show that LiFeSb in the LiFeAs structure would be dynamically stable in the sense of having no unstable phonon modes, and would have very similar electronic and magnetic properties to the layered Fe based superconductors. Importantly, a very similar structure for the Fermi surface and a spin density wave related to but stronger than that in the corresponding As compound is found. These results are indicative of possible superconductivity analogous to the Fe-As based compounds if the spin density wave can be suppressed by doping or other means. Prospects for synthesizing this material in pure form or in solid solution with FeTe are discussed.

PACS numbers: 74.25.Jb, 74.70.Dd, 71.18.+y, 74.25.Kc

The finding of high temperature superconductivity ($T_c \sim 26\text{K}$) in electron-doped $\text{LaFeAsO}_{1-x}\text{F}_x$,¹ has resulted in widespread interest and exploration of related materials, some of which have T_c exceeding 55K. In particular, superconductivity has been found in iron based oxy-arsenides by replacing La with other rare-earth metals,^{2,3,4,5,6} as well as oxygen-free arsenides such as doped BaFe_2As_2 ,^{7,8} SrFe_2As_2 ,⁹ CaFe_2As_2 ,^{10,11} and LiFeAs .^{12,13,14} The common structural feature of this family of materials is the appearance of Fe-As layers. These consist of an Fe square planar sheet tetrahedrally coordinated by As atoms from above and below. In addition, superconductivity occurs in doped LaFePO ,^{15,16,17} although with a lower T_c and in PbO structure $\alpha\text{-FeSe}_{1-x}$.^{18,19,20} These latter compounds also feature an Fe square lattice and a tetrahedral coordination of the Fe, though not with As. Importantly, the critical temperature of FeSe_{1-x} increases strongly with either Te substitution¹⁹ or pressure, reaching 27 K.²⁰ This high value of T_c under pressure implies a relationship with the Fe-As superconductors, which is also supported by similarities of the properties and theoretical studies.²¹ At present there is strong interest in finding new high temperature Fe-based superconductors and especially in finding materials with higher critical temperature.

One obvious direction is to examine antimonides. This is motivated by the fact that the properties of LaFePO and LaFeAsO appear to be closely related, suggesting a similar mechanism of superconductivity, and furthermore the compound with the heavier pnictogen (As) has the higher T_c when doped. However, this is highly non-trivial from a chemical perspective because Sb has a strong tendency to form Sb-Sb bonds in compounds. This leads to a strong tendency for transition metal compounds of Sb to contain more Sb than transition metal, as for example in skutterudite CoSb_3 and $\text{LaFe}_4\text{Sb}_{12}$ or marcasite structure FeSb_2 , although FeSb is a known phase.²² One way forward is provided by noting the structural similarity of LiFeAs with PbO structure FeSe_{1-x} and FeTe_{1-x} .

The chalcogenides, whose chemical formulas should more correctly be written as Fe_{1+x}Se and Fe_{1+x}Te , occur in a tetragonal structure with spacegroup $P4/nmm$ similar to LiFeAs, and consist of an Fe square lattice tetrahedrally coordinated with Se/Te ions, the same as in the structure of the Fe-As superconductors.^{23,24,25,26} These chalcogenides form with excess Fe, which occurs in a partially filled $2c$ site, in particular the cation site forming formally an enlarged tetrahedron around the Fe and approximately five-fold coordinated by Te.^{23,25,26} This is the same site that is occupied by Li in LiFeAs. Therefore, there is a close structural similarity between LiFeAs and the chalcogenides Fe_{1+x}Se and Fe_{1+x}Te . In particular the structure of LiFeAs is obtained by allowing full filling of the $2c$ cation site with Li^+ and replacement of Te^{2-} by As^{3-} . Therefore we focus on hypothetical LiFeSb since it may be possible to form it, or at the very least some range of solid solution between Fe_{1+x}Te and LiFeSb should be experimentally accessible, especially considering that alloys of related phases containing Te and Sb typically form as in e.g. the Bi-Sb-Te, AgSbTe-PbTe and AgSbTe-GeTe thermoelectrics, and also that there are many known Zintl type phases based on Li, Sb and metal atoms.

The crystal structure of LiFeSb is assumed to be isostructural with LiFeAs with the space group of $P4/nmm$.^{12,13,14} As shown in Fig. 1, the Fe-Sb layers formed by edge-shared tetrahedral FeSb_4 units are alternately spaced along the c -axis direction, and intercalated with Li. Note that the Li, which occurs as Li^+ , is coordinated by Sb. The structural parameters were calculated by local density approximation (LDA) total energy minimization with the full-potential linearized augmented plane wave (LAPW) method.²⁷ The calculated tetragonal lattice parameters are $a = 4.0351 \text{ \AA}$, $c = 6.3712 \text{ \AA}$, and internal coordinates $\text{Li}(2c)$ (0.25,0.25,0.697), $\text{Fe}(2a)$ (0.75,0.25,0), $\text{Sb}(2c)$ (0.25,0.25,0.228). The Fe-Sb bond length is 2.486 \AA , slightly larger than 2.4204 \AA for LiFeAs,¹⁴ which might be attributed to the larger size

of the Sb^{3-} anion relative to As^{3-} . The Fe-Fe distance is 2.853\AA , also a bit larger than the corresponding value (2.6809\AA) in LiFeAs , but still short enough for direct Fe-Fe interaction.

The electronic structure and magnetic properties calculations were performed within LDA-LAPW method, similar to previous reports.^{21,28,29} LAPW sphere radii of $1.8 a_0$, $2.0 a_0$, and $2.1 a_0$ were used for Li, Fe and Sb, respectively. Converged basis sets were used. These consisted of LAPW functions with a planewave cutoff determined as $R_{\text{Li}}k_{\text{max}}=8.0$ plus local orbitals both to relax linearization and to include the semicore states. The zone sampling for the self-consistent calculations was done using the special \mathbf{k} -points method, with a $16 \times 16 \times 8$ grid. Finer grids were used for the density of states and Fermi surface. The lattice dynamical properties were calculated through the frozen phonon method³⁰ (or small displacement method³¹). The required forces were obtained through the projector augmented-wave (PAW) method³² in VASP code, within the generalized gradient approximation of Perdew, Burke and Ernzerhof (PBE).³³ We also fully relaxed crystal structure and calculated electronic structures with PBE-PAW method and the results show excellent agreement with those by LDA-LAPW method (with a remarkably small maximum discrepancy of 0.3% in structural parameters). This cross-checking supports the reliability of the calculations and consistency of the different methods employed. We emphasize that these calculations for the structure were done without the inclusion of magnetism.

As discussed in detail in Ref. 34, there is generally a substantial underestimation of the pnictogen heights in these compounds when calculations are done in this way, while on the other hand in magnetic calculations LDA and GGA results differ, with the GGA giving much larger magnetic moments than experiment as well as magnetism that persists throughout the phase diagram in disagreement with experiment. LDA calculations done with the GGA structure give an intermediate state less magnetic than the GGA calculations but more magnetic than experiment, while in magnetic LDA calculations the As height is still substantially underestimated. In fact, LDA calculations done at the non-magnetic LDA As height (which agrees with the GGA As height) give the weakest magnetism, closest to experiment, though still overestimating the strength of the SDW. It was conjectured that these problems are consequences of strong spin fluctuations.

This conjecture is supported both by comparison of theoretical results with experiment as well as experimental observations, such as the highly unusual increasing with T susceptibility, $\chi(T)$ above the spin density wave ordering temperature observed in some compounds.^{35,36,37} Such an increasing shape suggests that strong magnetic correlations with the character of the SDW persist well above T_N ; since this itinerant magnetic state is driven by electrons at the Fermi energy a related electronic reconstruction of the Fermi surface may

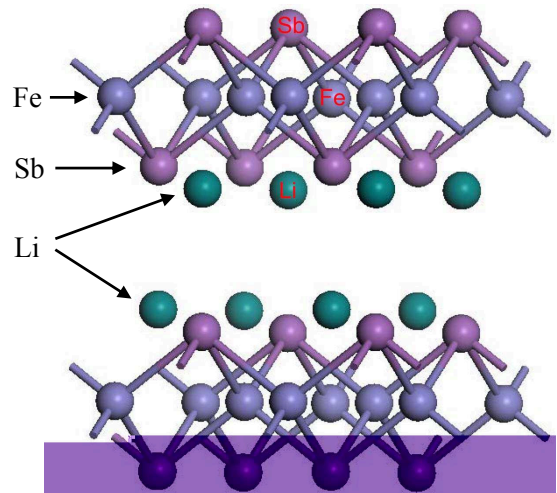


FIG. 1: (Color online) Crystal structure of hypothetical LiFeSb with the relaxed structural parameters from LAPW-LDA total energy minimization. Note that while for clarity similar size spheres are used for the different atoms, from a crystal chemical point of view Sb^{3-} anions are very large while Li^+ is very small.

also be expected above T_N ; this may be seen perhaps in photoemission. This shape of $\chi(T)$ persists also in the normal state for doped samples, where there is superconductivity but no SDW.³⁷ Other evidence for strong spin fluctuations comes from core level spectroscopy³⁸ and transport data showing strong scattering above the ordering temperature.³⁵ Returning to the increasing $\chi(T)$, which continues up to high temperature, one interesting possibility is that this represents preformed pairs that can condense into either superconducting or SDW order and which begin forming at a very high non-observed temperature above which $\chi(T)$ would return to a more normal decreasing with T shape. In any case, in this work where we compare the compounds, we consistently used the relaxed atomic coordinates from non-magnetic calculations.

A requirement for a compound to be made is that the lattice be stable. We verified that this is the case for hypothetical LiFeSb by calculating the vibrational modes of the compound.³⁹ We find no soft or unstable modes and no soft elastic constants. The calculated phonon dispersion curve and phonon DOS for LiFeSb are shown in Fig. 2. Due to larger difference in atomic weights compared to LiFeAs the phonon spectrum of LiFeSb is divided into three separated manifolds. The region of high frequencies (above 275 cm^{-1}) is dominated by Li, while the moderate (between 200 and 275 cm^{-1}) and low (below 150 cm^{-1}) frequency manifolds mainly derive from Fe and Sb respectively. As may be seen, all the phonon frequencies are safely positive and there are no optical phonon branches with dispersions that dip towards zero frequency. This shows that the $P4/nmm$ structure of LiFeSb is dynamically stable. We also calculated the heat of formation from the elements. We obtain -0.51 eV

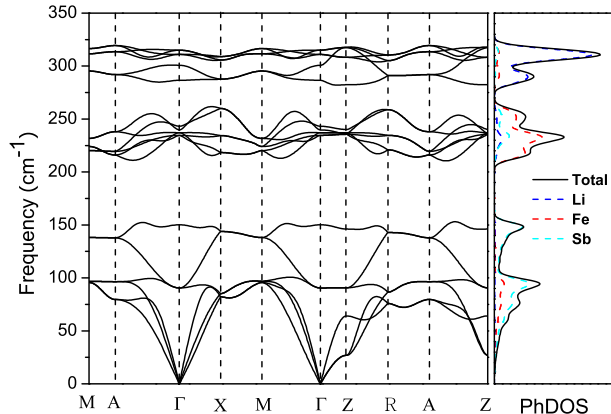


FIG. 2: Left panel: Calculated phonon dispersion curves for LiFeSb. Right panel: (Color online) The total and projected (onto atoms) phonon DOS.

per formula unit (i.e. -49 kJ/mole formula unit), which indicates that the compound may be delicate but would at least be stable against decomposition into elements. Therefore we continue to discuss the magnetic and electronic properties.

Our main results for the electronic structure of LiFeSb, are given in Figs. 3, 4 and 5, which show the calculated band structure, electronic density of states (DOS), and Fermi surface, respectively. The general shape of band structure near the Fermi energy E_F is very similar to the calculated results for LiFeAs.^{29,40} There are compensating heavy hole and electron Fermi surfaces, with two electron cylinders at the zone corner (M) and hole surfaces around the zone center. The hole surfaces consist of 2D cylindrical and small heavy 3D sections. Similar to the Fe-As based materials,^{28,29,40,41,42,43} the electron Fermi surface of LiFeSb may be described as two intersecting cylindrical sections of elliptical cross-section, with major axes at 90° to each other and centered at the M point. We find somewhat a different hole Fermi surface structure from LiFeAs, with only one complete hole cylinder at the zone center, along with two additional heavier 3D hole pockets. It can be seen that electron cylinders are more two dimensional than in LiFeAs. Also, the 2D hole cylinder is close in size to that of the electron cylinders, which may be expected to lead to nesting. Thus there is strong nesting of Fermi surface at the two-dimensional (2D) nesting vector (π, π) . This would be expected to lead to an SDW state related to the M point, as in the Fe-As based superconductors.^{42,43,44,45,46,47} We studied the energetic stability of SDW state for LiFeSb directly using a doubled cell containing lines of Fe atoms with parallel spin in the Fe-Sb layers and do in fact find a stable SDW state. Within the LDA with the LDA structural parameters, the local spin moment of the SDW state is $1.12 \mu_B$, much larger than the corresponding value (0.69

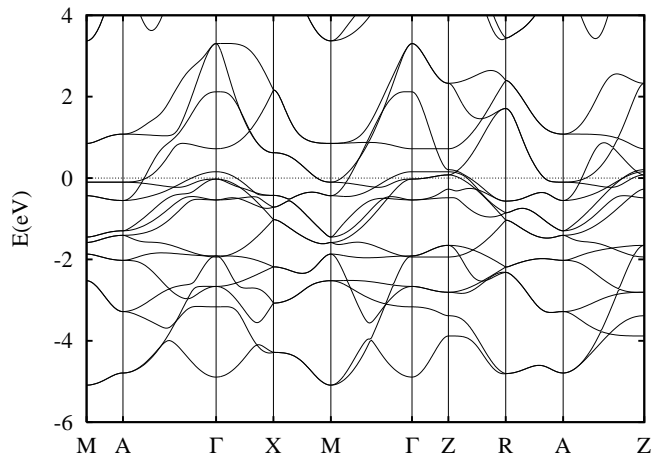


FIG. 3: Calculated LDA band structure of LiFeSb using the calculated structural parameters. The Fermi energy is at 0 eV.

μ_B) for LiFeAs calculated in the same way,²⁹ indicating that LiFeSb has a more stable SDW.

The qualitative similarity to the electronic structure of the Fe-As based superconductors^{28,29,40,41,42,43} is also evident in the DOS. The Sb p states are located mainly below -1.7 eV relative to E_F , and are only moderately hybridized with the Fe d states, indicating that the Sb is anionic with valence close to 3. The DOS near the Fermi level is dominated by Fe d states, deriving from the metallic Fe^{2+} sublattice with direct Fe-Fe interactions, and has a characteristic pseudogap near E_F . In fact, E_F lies on the low energy side of the pseudogap, where $N(E_F)$ is decreasing with energy but still high. Specifically, the value of $N(E_F) = 2.2 \text{ eV}^{-1}$ per Fe both spins is much larger than that for LiFeAs and is comparable to the oxy-arsenides (e.g. $N(E_F)$ calculated in the same way for LaFeAsO is 2.6 eV^{-1}), which are the Fe-As compounds with higher T_c . For comparison, the values for LiFeAs and BaFe_2As_2 are 1.79 eV^{-1} and 1.53 eV^{-1} on a per Fe basis, respectively, when calculated in the same way.²⁹

Within the Stoner theory, the appearance of an instability of the paramagnetic state towards itinerant ferromagnetism would be determined by the criterion $N(E_F)I > 1$, where I is the Stoner parameter, with the typical value in Fe compounds of $I \sim 0.7 - 0.8 \text{ eV}$. Thus, the significantly higher $N(E_F)$ in LiFeSb would inevitably place it closer to magnetism in general than LiFeAs or BaFe_2As_2 . While the mechanism of superconductivity has yet to be established, there is accumulating evidence of a connection with magnetism, and so chemically tuning the proximity to magnetism is a likely strategy for modifying the superconductivity. In general, the Fe-based superconductors exhibit temperature-induced magnetic and structural phase transitions with a spin density wave (SDW) character related to the Fermi surface nesting.^{7,48,49} Superconductivity appears as the

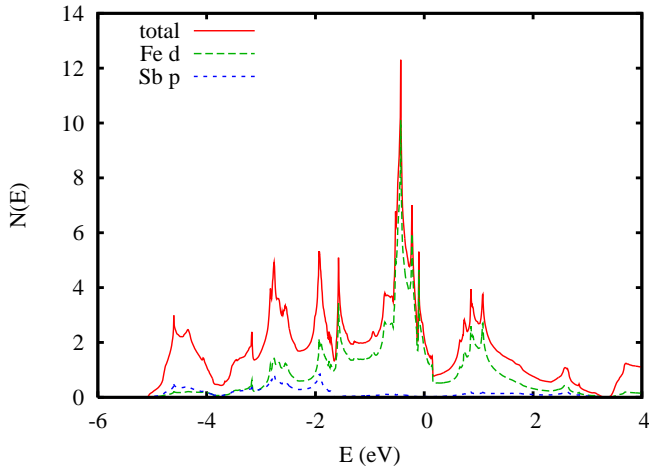


FIG. 4: (Color online) Calculated total and partial electronic DOS for LiFeSb, on a per formula unit basis. The contribution from Li-2s state lying in deep energy range was not shown. The projections are onto the LAPW spheres, thus the Sb-5p was slightly underestimated owing to its more extended orbitals.

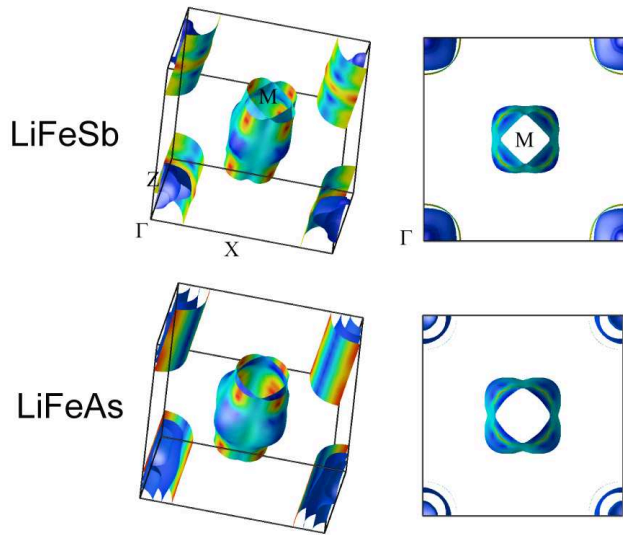


FIG. 5: (Color online) Calculated LDA Fermi surface of LiFeSb in comparison with LiFeAs, shaded by band velocity with blue as low velocity. The right panels are top views along the c -axis direction.

spin density wave is suppressed by doping or pressure.

Electronic structure calculations^{28,29,40,41,42,43} show that all these materials have compensating small electron and hole Fermi surfaces, with nesting between 2D electron sheets and heavier 2D hole sheets, which are separated by (π, π) . This is associated with the SDW magnetic state.^{42,43,44,45,46,47}

Within this framework, the idea that going to heavier ligands may be beneficial for superconductivity is supported by previous density functional calculations. These

have shown that the electronic structures of the Fe-As superconductors are rather ionic with the exception of the Fe layers, which are metallic due mainly to Fe-Fe interactions.²⁸ This is different from the cuprates where hopping is through the O atoms in the CuO_2 planes, and implies that the ligand (O/As) atoms play a less crucial role in the properties of the FeAs superconductors than in the cuprates. Furthermore, it has been found that there is a strong connection between the As position above the Fe plane and the magnetic properties, with higher positions yielding stronger magnetism.^{34,44} This is supported by calculations comparing FeSe and FeTe²¹ and for hypothetical LaFeSbO in comparison with LaFeAsO.⁵⁰ In both cases stronger magnetism is found in going to the larger ligand, which because of its size is then further from the Fe plane yielding narrower bands and higher $N(E_F)$. The combination of a stronger SDW and higher $N(E_F)$ leading to stronger spin fluctuations in general and in particular away from the nesting vector may be crucial. This is because the ordered SDW is antagonistic to superconductivity. In scenarios where the associated spin-fluctuations that couple the electron and hole Fermi surface sections play the main role in pairing, as discussed in Refs. 46 and 51, spin fluctuations away from the nesting vector while not directly pairing may play a very important role. This is because they would compete with the SDW preventing long range order and leading to a renormalized paramagnetic state even though in mean field the SDW may be the predicted ground state as in the LDA. In any case, in these scenarios the role of doping is to weaken and broaden the peak in the susceptibility associated with the nesting, destroying the SDW in favor of a state with spin fluctuations around the zone corner. We also note that in case the SDW is not destroyed by doping alone, it may be possible for it to be destroyed by disorder yielding superconductivity in an alloy system such as $\text{Fe}_{1+x}\text{Te} - \text{LiFeSb}$. This may be possible because the SDW is related to a divergence in the peak of the susceptibility, $\chi(\mathbf{q})$, while within a spin-fluctuation mediated framework superconductivity will in general be related not to the peak value but to an integral over the Fermi surface, i.e. a high average value over some region of the zone, specifically the region for which the wavevector connects the electron and hole Fermi surfaces.^{46,52} Also near divergences will be pair breaking for superconductivity. This may also partly explain why superconductivity in these phases is relatively robust against alloying with Zn or Co in the Fe planes,^{53,54,55} even though in an unconventional superconductor scattering, including non-magnetic scattering, is pair breaking. Thus, even if the magnetic ground state cannot be destroyed by doping in Fe_{1+x}Te or LiFeSb (supposing that this can be synthesized) it may be destroyed in favor of superconductivity in the solid solution between these two compounds.

In any case, our results show that, if it can be synthesized, LiFeSb will have electronic and magnetic properties closely related to those of the Fe-As based superconductors, and in particular will show a qualitatively

similar Fermi surface structure and tendency towards an SDW state. In comparison with LiFeAs, it will have a higher $N(E_F)$ and a stronger SDW. This may favor higher critical temperatures, at least within a scenario with interband pairing mediated by spin fluctuations associated with the Fermi surface nesting. In addition, this material is found to be dynamically stable, evidenced by the absence of any unstable phonon modes. It is also worth noting that this solid solution contains no elements as toxic as As. As such it would be of considerable interest to attempt synthesis of this compound or its solid

solution with Fe_{1+x}Se , LiFeAs and especially Fe_{1+x}Te .

Acknowledgments

We are grateful for helpful discussions with D. Mandrus, I.I. Mazin and B.C. Sales. This work was supported by the Department of Energy, Division of Materials Sciences and Engineering.

-
- ¹ Y. Kamihara, T. Watanabe, M. Hirano, and H. Hosono, J. Am. Chem. Soc. **130**, 3296 (2008).
 - ² C. Wang, L. Li, S. Chi, Z. Zhu, Z. Ren, Y. Li, Y. Wang, X. Lin, Y. Luo, S. Jiang, et al., Europhys. Lett. **83**, 67006 (2008).
 - ³ Z.-A. Ren, G.-C. Che, X.-L. Dong, J. Yang, W. Lu, W. Yi, X.-L. Shen, Z.-C. Li, L.-L. Sun, F. Zhou, et al., Europhys. Lett. **83**, 17002 (2008).
 - ⁴ A. S. Sefat, M. A. McGuire, B. C. Sales, R. Jin, J. Y. Howe, and D. Mandrus, Phys. Rev. B **77**, 174503 (2008).
 - ⁵ G. F. Chen, Z. Li, D. Wu, G. Li, W. Z. Hu, J. Dong, P. Zheng, J. L. Luo, and N. L. Wang, Phys. Rev. Lett. **100**, 247002 (2008).
 - ⁶ H. Wen, G. Mu, L. Fang, H. Yang, and X. Zhu, Europhys. Lett. **82**, 17009 (2008).
 - ⁷ M. Rotter, M. Tegel, I. Schellenberg, W. Hermes, R. Pottgen, and D. Johrendt, Phys. Rev. B **78**, 020503 (2008).
 - ⁸ M. Rotter, M. Tegel, and D. Johrendt, Phys. Rev. Lett. **101**, 107006 (2008).
 - ⁹ G. F. Chen, Z. Li, G. Li, W. Z. Hu, J. Dong, X. D. Zhang, P. Zheng, N. L. Wang, and J. L. Luo, Chin. Phys. Lett. **25**, 3403 (2008).
 - ¹⁰ N. Ni, S. Nandi, A. Kreyssig, A. I. Goldman, E. D. Mun, S. L. Bud'ko, and P. C. Canfield, Phys. Rev. B **78**, 014523 (2008).
 - ¹¹ M. S. Torikachvili, S. L. Bud'ko, N. Ni, and P. C. Canfield, Phys. Rev. Lett. **101**, 057006 (2008).
 - ¹² G. Wu, H. Chen, Y. L. Xie, Y. J. Yan, T. Wu, R. H. Liu, X. F. Wang, D. F. Fang, J. J. Ying, and X. H. Chen, Phys. Rev. B **78**, 092503 (2008).
 - ¹³ M. J. Pitcher, D. R. Parker, P. Adamson, S. J. C. Herkelrath, A. T. Boothroyd, and S. J. Clarke, arXiv:0807.2228 (2008).
 - ¹⁴ J. H. Tapp, Z. Tang, B. Lv, K. Sasmal, B. Lorenz, P. C. W. Chu, and A. M. Guloy, Phys. Rev. B **78**, 060505(R) (2008).
 - ¹⁵ Y. Kamihara, H. Hiramatsu, M. Hirano, R. Kawamura, H. Yanagi, T. Kamiya, and H. Hosono, J. Am. Chem. Soc. **128**, 10012 (2006).
 - ¹⁶ C. Y. Liang, R. C. Che, H. X. Yang, H. F. Tian, R. J. Xiao, J. B. Lu, R. Li, and J. Q. Li, Supercond. Sci. and Technol. **20**, 687 (2007).
 - ¹⁷ T. M. McQueen, M. Regulacio, A. J. Williams, Q. Huang, J. W. Lynn, Y. S. Hor, D. V. West, M. A. Green, and R. J. Cava, Phys. Rev. B **78**, 024521 (2008).
 - ¹⁸ F.-C. Hsu, J.-Y. Luo, K.-W. Yeh, T.-K. Chen, T.-W. Huang, P. M. Wu, Y.-C. Lee, Y.-L. Huang, Y.-Y. Chu, D.-C. Yan, et al., Proc. Nat. Acad. Sci (USA) **105**, 14262 (2008).
 - ¹⁹ K.-W. Yeh, T.-W. Huang, Y.-L. Huang, T.-K. Chen, F.-C. Hsu, P. M. Wu, Y.-C. Lee, Y.-Y. Chu, C.-L. Chen, J.-Y. Luo, et al., arXiv:0808.0474 (2008).
 - ²⁰ Y. Mizuguchi, F. Tomioka, S. Tsuda, T. Yamaguchi, and Y. Takano, Appl. Phys. Lett. **93**, 152505 (2008).
 - ²¹ A. Subedi, L. Zhang, D. J. Singh, and M.-H. Du, Phys. Rev. B **78**, 134514 (2008).
 - ²² A. Kjekshus and K. P. Walseth, Acta Chemica Scandinavica **23**, 2621 (1969).
 - ²³ F. Gronvold and H. Haraldsen, Acta Chem. Scand **8**, 1927 (1954).
 - ²⁴ S. Chiba, J. Phys. Soc. Japan **10**, 837 (1955).
 - ²⁵ D. M. Finlayson, D. Greig, J. P. Llewellyn, and T. Smith, Proc. Phys. Soc. B **69**, 860 (1956).
 - ²⁶ J. Leciejewicz, Acta Chem. Scand **17**, 2593 (1963).
 - ²⁷ D. J. Singh and L. Nordstrom, *Plane waves Pseudopotentials and the LAPW Method, 2nd Edition* (Springer, Berlin, 2006).
 - ²⁸ D. J. Singh and M.-H. Du, Phys. Rev. Lett. **100**, 237003 (2008).
 - ²⁹ D. J. Singh, Phys. Rev. B **78**, 094511 (2008).
 - ³⁰ A. Togo, *fropho: Frozen Phonon analyzer for periodic boundary condition materials*.
 - ³¹ D. Alfé, G. D. Price, and M. J. Gillan, Phys. Rev. B **64**, 045123 (2001).
 - ³² G. Kresse and D. Joubert, Phys. Rev. B **59**, 1758 (1999).
 - ³³ J. Perdew, K. Burke, and M. Ernzerhof, Phys. Rev. Lett. **77**, 3865 (1996).
 - ³⁴ I. I. Mazin, M. D. Johannes, L. Boeri, K. Koepernik, and D. J. Singh, Phys. Rev. B **78**, 085104 (2008).
 - ³⁵ M. A. McGuire, A. D. Christianson, A. S. Sefat, B. C. Sales, M. D. Lumsden, R. Jin, E. A. Payzant, D. Mandrus, Y. Luan, V. Keppens, et al., Phys. Rev. B **78**, 094517 (2008).
 - ³⁶ H. H. Klauss, H. Luetkens, R. Klingeler, C. Hess, F. J. Litterst, M. Kraken, M. M. Korshunov, I. Eremin, S. L. Drechsler, R. Khasanov, et al., Phys. Rev. Lett. **101**, 077005 (2008).
 - ³⁷ R. Klingeler, N. Leps, I. Hellmann, A. Popa, A. Kondrat, J. Hamann-Borrero, G. Behr, V. Kataev, and B. Buechner, arXiv:0808.0708 (2008).
 - ³⁸ F. Bondino, E. Magnano, F. Parmigiani, M. A. McGuire, A. S. Sefat, B. C. Sales, R. Jin, D. Mandrus, E. W. Plummer, D. J. Singh, et al. (????).
 - ³⁹ S. Baroni, S. de Gironcoli, A. Dal Corso, and P. Giannozzi,

- Rev. Mod. Phys. **73**, 515 (2001).
- ⁴⁰ I. A. Nekrasov, Z. V. Pchelkina, and M. V. Sadovskii, arXiv:0807.1010 (2008).
- ⁴¹ I. A. Nekrasov, Z. V. Pchelkina, and M. V. Sadovskii, JETP Lett. **88**, 144 (2008).
- ⁴² F. Ma and Z.-Y. Lu, Phys. Rev. B **78**, 033111 (2008).
- ⁴³ F. Ma, Z.-Y. Lu, and T. Xiang, arXiv:0806.3526 (2008).
- ⁴⁴ Z. P. Yin, S. Lebegue, M. J. Han, B. Neal, S. Y. Savrasov, and W. E. Pickett, Phys. Rev. Lett. **101**, 047001 (2008).
- ⁴⁵ J. Dong, H. J. Zhang, G. Xu, Z. Li, G. Li, W. Z. Hu, D. Wu, G. F. Chen, X. Dai, J. L. Luo, et al., Europhys. Lett. **83**, 27006 (2008).
- ⁴⁶ I. I. Mazin, D. J. Singh, M. D. Johannes, and M.-H. Du, Phys. Rev. Lett. **101**, 057003 (2008).
- ⁴⁷ T. Yildirim, Phys. Rev. Lett. **101**, 057010 (2008).
- ⁴⁸ C. de la Cruz, Q. Huang, J. W. Lynn, J. Li, W. Ratcliff II, J. L. Zarestsky, H. A. Mook, G. F. Chen, J. L. Luo, N. L. Wang, et al., Nature **453**, 899 (2008).
- ⁴⁹ T. Nomura, S. W. Kim, Y. Kamihara, M. Hirano, P. V. Sushko, K. Kato, M. Takata, A. L. Shluger, and H. Hosono, arXiv:0804.3569 (2008).
- ⁵⁰ C.-Y. Moon, S. Y. Park, and H. J. Choi, arXiv:0808.1348 (2008).
- ⁵¹ K. Kuroki, S. Onari, R. Arita, H. Usui, Y. Tanaka, H. Kotani, and H. Aoki, Phys. Rev. Lett. **101**, 087004 (2008).
- ⁵² D. Fay and J. Appel, Phys. Rev. B **22**, 3173 (1980).
- ⁵³ A. S. Sefat, A. Huq, M. A. McGuire, R. Jin, B. C. Sales, and D. Mandrus, Phys. Rev. B **78**, 104505 (2008).
- ⁵⁴ A. S. Sefat, R. Jin, M. A. McGuire, B. C. Sales, D. J. Singh, and D. Mandrus, Phys. Rev. Lett. **101**, 117004 (2008).
- ⁵⁵ Y. K. Li, X. Lin, C. Wang, L. J. Li, Z. W. Zhu, M. H. Q. Tao, Q. B. Wang, G. H. Cao, and Z. A. Xu, arXiv:0808.0328 (2008).